# Effect of thickness on elastic-property measurements of thin films using atomic force acoustic microscopy

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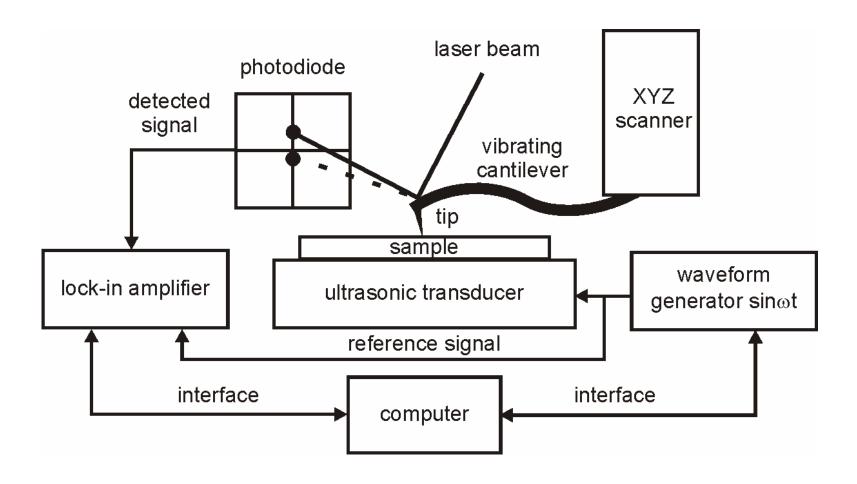


#### **Overview**

- Basics of AFAM
- •Characterization of the investigated thin-film samples; measurements of thickness, microstructure and texture
- Experimental results
- •Stress field in AFAM experiments.
  When can we neglect the influence of a substrate?
- Conclusions



# **Experimental set-up**



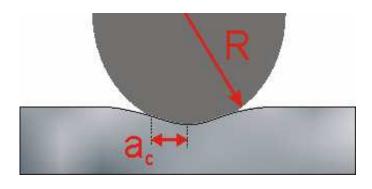
### **Experimental procedure**

cantilever: applied force: calibration:

 $k_c = 44 \text{ N/m}$  880 nN - 2640 nN, reference sample: single-crystal nickel

 $f_1 = 168 \text{ kHz}$  measured:  $f_{1con}$ ,  $f_{2con}$ , orientation: <100>

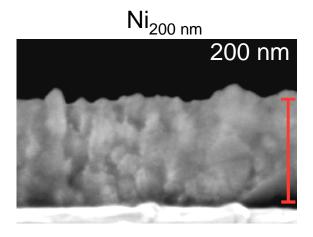
 $f_2 = 1042 \text{ kHz}$  calculated:  $k^*$  indentation modulus:  $M_{<100>} = 219 \pm 2 \text{ GPa}$ 

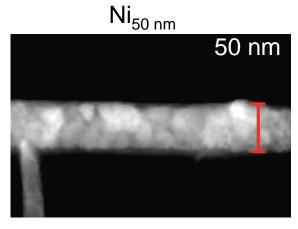


$$\mathsf{E}_{\mathsf{sam}}^* = \mathsf{E}_{\mathsf{ref}}^* \left( \frac{\mathsf{k}_{\mathsf{sam}}^*}{\mathsf{k}_{\mathsf{ref}}^*} \right)^\mathsf{n} \quad \left\{ \begin{array}{l} \mathsf{n} = \mathsf{1} \; \mathsf{for} \; \mathsf{a} \; \mathsf{flat} \; \mathsf{punch} \\ \mathsf{n} = \mathsf{3/2} \; \; \mathsf{for} \; \mathsf{a} \; \mathsf{hemisphere} \end{array} \right. \quad \frac{\mathsf{1}}{\mathsf{E}^*} = \frac{\mathsf{1}}{\mathsf{M}_{\mathsf{tip}}} + \frac{\mathsf{1}}{\mathsf{M}_{\mathsf{sample}}}$$

# **Cross-sectional SEM images of thin-film nickel samples**

Ni<sub>800 nm</sub>
800 nm





Measured thickness:

 $772 \pm 5 \text{ nm}$ 

 $204 \pm 4 \text{ nm}$ 

 $53 \pm 2 \text{ nm}$ 

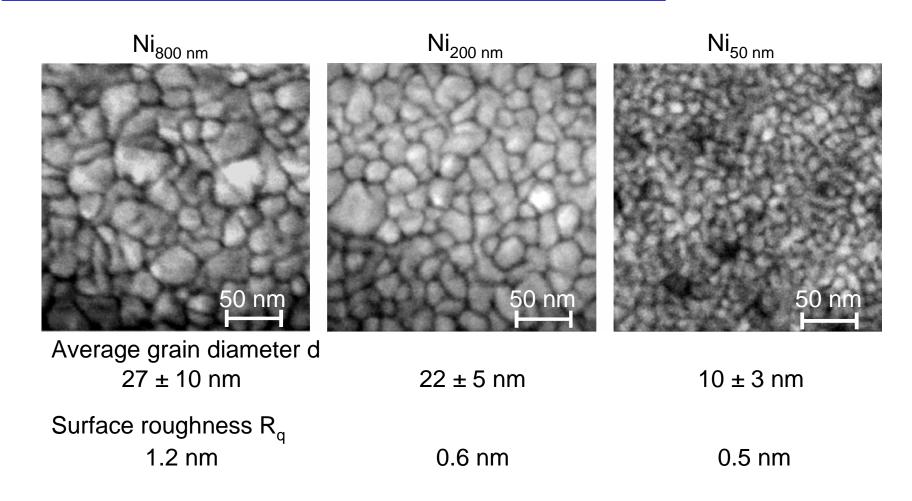
Method: DC magnetron sputtering

Substrate: <100> silicon

Preparation: W.H. Rippard and S.E. Russek, NIST Boulder



# Plane view SEM images of thin-film nickel samples



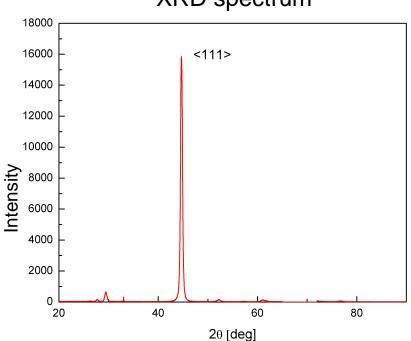


# **Texture study for 800 nm film**

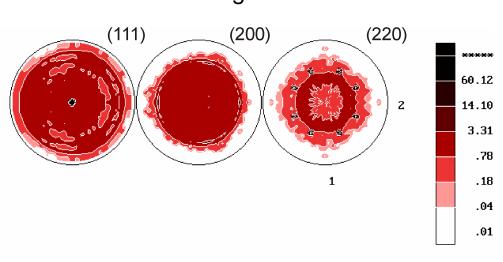
Ni is elastically anisotropic

Literature values for single-crystal Ni

XRD spectrum



 $M_{<100>}$  = 220 GPa  $M_{<111>}$  = 250 GPa Pole figures

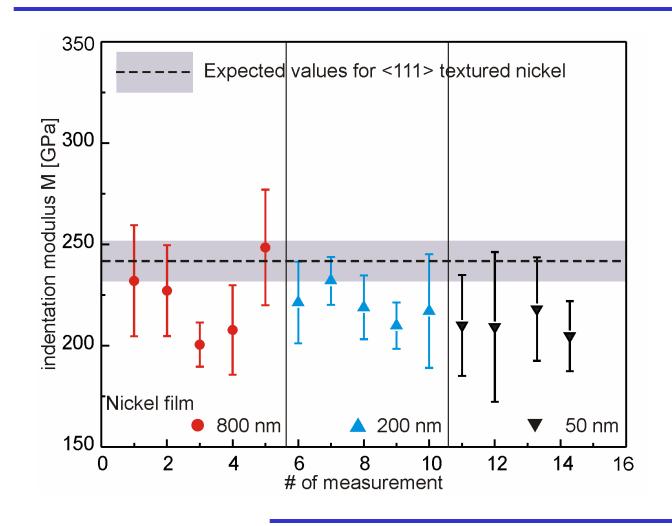


Literature value for <111> textured material

 $M = 242 \pm 10 GPa$ 



#### **AFAM** results for indentation modulus M



Average values of M

Ni<sub>800nm</sub> 223 ± 28 GPa

Ni<sub>200nm</sub> 220 ± 19 GPa

 $Ni_{50nm}$  210 ± 26 GPa

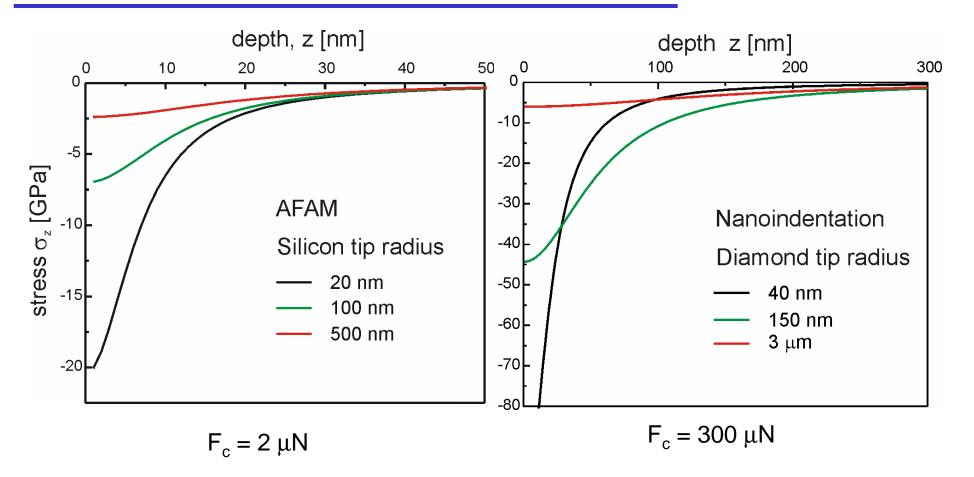
#### What can cause a reduction in values of elastic modulus?

• Large deformation of substrate at the interface of thin-film sample?

• Presence of closed pores (voids) and elemental impurities in the film?

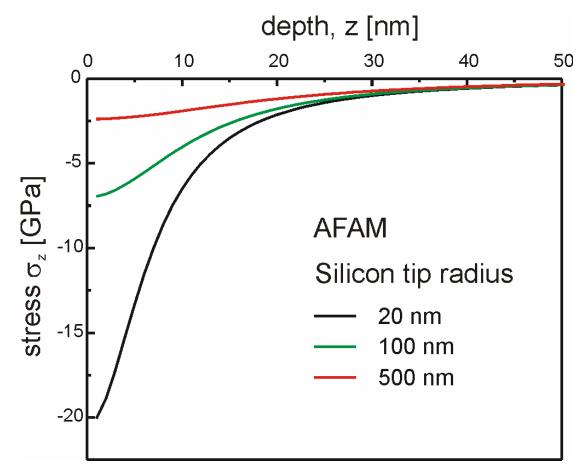
Change of elastic properties due to high intercrystalline volume content?

#### Stress field in AFAM and nanoindentation



Stress fields in AFAM are smaller and decrease faster than in nanoindentation

#### **Substrate influence**



Stress field under a spherical indenter:

enter: 
$$\sigma_z = -p_{\text{max}} \left(1 + \frac{z^2}{a_c^2}\right)^{-1}$$
 
$$p_{\text{max}} = \frac{3F_C}{2\pi a_c^2}$$
 
$$a_c = \sqrt[3]{\frac{3RF_C}{4E^*}}$$

Deformation  $\delta$  caused by the stress:

$$\delta = -\sigma_z \frac{\pi a_c}{2E^*}$$

#### Deformation of the substrate at the interface

R: 40 nm - 110 nm,  $\max F_c = 2640 \text{ nN}$ 

 $\delta_{_{\! S}}$  - deformation at the surface

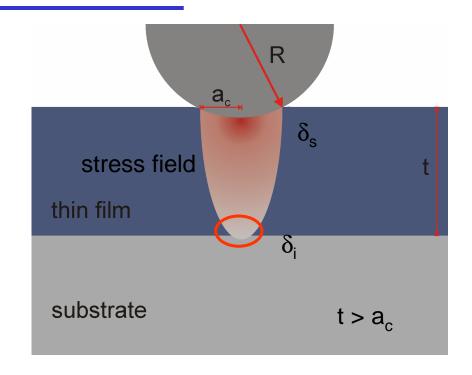
 $\delta_{i}$  - deformation at the interface

We can neglect the substrate influence if

$$\delta_s \gg \delta_i$$

We estimated that:

t (nm)	$\delta_{\rm s}$ (nm)	$\delta_{i}$ (nm)
800	~2	« 0.01
200	~2	« 0.01
50	~2	~ 0.1

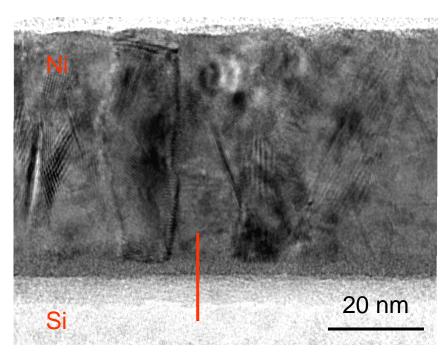


The values of M measured in our AFAM experiments were **not** influenced by the substrate!



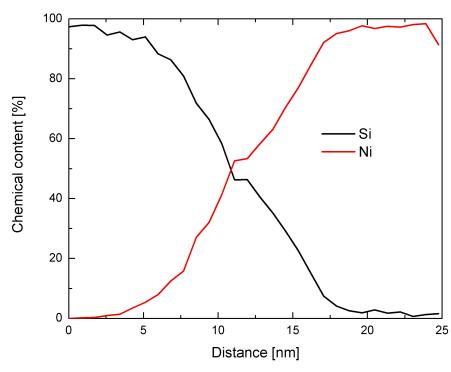
# What about closed porosity (voids) and impurities?

Neither SEM or TEM images showed closed porosity (voids)



line scan

Analysis of the EDS spectrum from the sample showed only Si and Ni and no other significant elemental impurities





#### **Grain size effects?**

 $Ni_{800nm}$   $Ni_{200nm}$   $Ni_{50nm}$ 

Grain size:  $27 \pm 10 \text{ nm}$   $22 \pm 5 \text{ nm}$   $10 \pm 3 \text{ nm}$ 

Polycrystalline materials consist of crystalline and intercrystalline (grain boundaries, triple junctions) components.

Crystalline component: 89 % 87 % 73 %

For nanocrystalline materials the effective elastic properties change with decreasing crystalline component

M<sub>eff</sub>: 228 GPa 223 GPa 202 GPa

 $M_{AFAM}$ : 223 ± 28 GPa 220 ± 19 GPa 210 ± 26 GPa

The reduction in M observed in our experiments is caused by the increasing volume of the intercrystalline components.



## **Summary**

We measured the elastic properties of ultra-thin nickel film samples with AFAM.

The values of M we obtained were lower than those expected for a textured material.

Possible causes for the observed reduction in M:

The substrate influence on the measured values: NO

Presence of porosity, impurities, ....: NO

Grain size effect: YES

AFAM offers the flexibility to measure very thin films. The minimum thickness depends on the relative elastic properties, tip radius and applied loads.



#### Poster session, Wednesday evening, December 1

# Influence of tip wear on atomic force acoustic microscopy experiments O10.16

